

## Crystal structure of $Zr_5AlGe_3$

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The new ternary intermetallic compound  $Zr_5AlGe_3$  was synthesized and its crystal structure was determined by X-ray single-crystal diffraction (structure type  $Hf_5CuSn_3$ , Pearson symbol  $hP18$ , space group  $P6_3/mcm$ ,  $a = 8.104(3)$ ,  $c = 5.654(2)$  Å,  $Z = 2$ ). The structure type  $Hf_5CuSn_3$  is a ternary variant of the structure type  $Ti_5Ga_4$  and a filled-up variant of the  $Mn_5Si_3$  type.  $ZrGe_6$  and  $AlZr_6$  octahedra are connected by common faces and form 1D-rods along the crystallographic direction [001], the former sharing edges to build up a 3D-framework with channels hosting the latter. At 600°C no solid solution was observed between the  $Mn_5Si_3$ -type binary compound  $Zr_5Ge_3$  and the ternary compound  $Zr_5AlGe_3$ .

Zirconium / Aluminum / Germanium / X-ray single-crystal diffraction / Crystal structure

### Introduction

The phase diagram of the ternary system Zr–Al–Ge has to our knowledge not yet been investigated. The structural evolution in the Al-rich alloys  $ZrAl_{3-x}Ge_x$  ( $x = 0-0.8$ ) at 600°C, when Al atoms are progressively replaced by Ge atoms, and the crystal structure of the ternary compound  $ZrAl_{2.5}Ge_{0.5}$  (structure type  $TiAl_3$ , Pearson symbol  $tI8$ , space group  $I4/mmm$ :  $a = 3.92395$ ,  $c = 9.0476$  Å) were reported in [1]. Information on the phase relations in the Zr-rich part of the system Zr–Al–Ge is absent in the literature.

The aim of the present work was to refine the crystal structure of the new ternary compound  $Zr_5AlGe_3$ , discovered during an investigation of the Zr–Al–Ge system. The structure was identified as belonging to the structure type  $Hf_5CuSn_3$  (Pearson symbol  $hP18$ , space group  $P6_3/mcm$ ) [2], which is a ternary filled-up variant of the  $Mn_5Si_3$  type ( $hP16$ ,  $P6_3/mcm$ ) [3]. Among the related ternary systems of Ti, Zr, or Hf with *p*-elements of group III, IV, or IV, isotypic ternary phases have only been reported for some of the systems {Zr,Hf}–{Al,Ga}–{Sn,Sb} (Table 1). However, in the binary systems Ti–{Ga,Sb}, Zr–{Al,Ga,Sn,Pb,Sb}, and Hf–Sn the existence of several compounds  $T_5M_4$  adopting the binary variant of the  $Hf_5CuSn_3$  type, the  $Ti_5Ga_4$  type ( $hP18$ ,  $P6_3/mcm$ ) [10], has been reported. Like

$Hf_5CuSn_3$ , the structure type  $Ti_5Ga_4$  is a filled-up derivative of the  $Mn_5Si_3$  type. In the system Zr–Al–Ge three binary compounds have been reported with one of these types:  $Zr_5Al_3$  ( $Mn_5Si_3$ ,  $hP16$ ,  $P6_3/mcm$ ,  $a = 8.184$ ,  $c = 5.702$  Å) [11],  $Zr_5Al_4$  ( $Ti_5Ga_4$ ,  $hP18$ ,  $P6_3/mcm$ ,  $a = 8.447$ ,  $c = 5.810$  Å) [12], and  $Zr_5Ge_3$  ( $Mn_5Si_3$ ,  $hP16$ ,  $P6_3/mcm$ ,  $a = 7.99$ ,  $c = 5.54$  Å) [13]. It should be noted that the voids in  $Mn_5Si_3$ -type structures can sometimes be occupied by small atoms, easily overlooked, such as C, N, or O (structure type  $Hf_5CuSn_3$ ), e.g.  $Zr_5Al_3C$ ,  $Zr_5Al_3N$ ,  $Zr_5Al_3O_{0.17}$  [4].

### Experimental

An alloy of nominal composition  $Zr_{55.6}Al_{11.1}Ge_{33.3}$  was synthesized from high-purity metals ( $\geq 99.9$  mass %) by arc melting in a water-cooled copper crucible under a purified argon atmosphere, using a tungsten electrode and Ti as a getter. To achieve homogeneity the sample was melted twice. After the synthesis the ingot was wrapped into tantalum foil, sealed in a quartz ampoule under vacuum and annealed at 600°C for 720 h. Finally the ampoule with the sample was quenched into cold water. The mass loss did not exceed 1 mass% of the total mass, which was approximately 1 g.

**Table 1** Unit-cell parameters for ternary compounds with  $Hf_5CuSn_3$ -type structures (Pearson symbol  $hP18$ , space group  $P6_3/mcm$ ) reported in the systems  $\{Zr,Hf\}-\{Al,Ga\}-\{Sn,Sb\}$  [4].

Compound	Unit-cell parameters, Å		Ref.
	<i>a</i>	<i>c</i>	
$Zr_5AlSn_3$	8.655	5.871	[5]
$Zr_5GaSn_3$	8.6599	5.8794	[5]
$Hf_5GaSn_3^a$	8.5564	5.7859	[6,7]
$Zr_5AlSb_3$	8.5802	5.8465	[8]
$Hf_5GaSb_3$	8.4747	5.7190	[9]

<sup>a</sup> limiting composition of the solid solution  $Hf_5Ga_xSn_3$  ( $x = 0-1$ ) of inclusion type

A prism-like single crystal was extracted from the alloy, mounted on a glass fiber and X-ray diffraction data were collected on a Rigaku AFC7 diffractometer, equipped with a Mercury CCD detector (graphite monochromator, Mo  $K\alpha$  radiation,  $\lambda = 0.71073$  Å) at room temperature. No significant variations of the intensities were observed during the data collection. An absorption correction was performed applying a multi-scan procedure.

The structure was solved by direct methods in the centrosymmetric hexagonal space group  $P6_3/mcm$ ; the positional and anisotropic displacement parameters were refined by the full-matrix least-squares method, using the program package WinCSD [14]. Experimental details and crystallographic data for  $Zr_5AlGe_3$  are listed in Table 2.

## Results and discussion

The structure of the ternary compound  $Zr_5AlGe_3$  belongs to the structure type  $Hf_5CuSn_3$  (Pearson symbol  $hP18$ , space group  $P6_3/mcm$ ), which

represents a ternary ordered variant of the structure type  $Ti_5Ga_4$ , which is a filled up derivative of the structure type  $Mn_5Si_3$  ( $hP16$ ,  $P6_3/mcm$ ). Atom coordinates and isotropic displacement parameters from the refinement are given in Table 3. The structure of  $Zr_5AlGe_3$  is characterized by two sites occupied by Zr atoms, one site by Al atoms, and one site by Ge atoms. The Al atoms occupy Wyckoff position  $2b$ , which is empty in the structure of the binary germanide  $Zr_5Ge_3$  (structure type  $Mn_5Si_3$ ). According to preliminary results of an investigation of the phase relations in the system Zr–Al–Ge, at 600°C  $Zr_5AlGe_3$  is a distinct ternary compound, whereas the binary compound  $Zr_5Ge_3$  dissolves up to 8 at.% Al forming a solid solution  $Zr_5Al_xGe_{3-x}$  ( $x = 0-0.64$ ) of the substitution type.

The interatomic distances and coordination numbers in the structure of  $Zr_5AlGe_3$  are summarized in Table 4. The interatomic distances correlate with the sums of the atomic radii of the elements ( $r_{Zr} = 1.60$ ,  $r_{Al} = 1.431$  and  $r_{Ge} = 1.225$  Å [15]) and are also in good agreement with the distances in the structures of known binary zirconium aluminides and germanides.

**Table 2** Experimental details and crystallographic data for  $Zr_5AlGe_3$ .

Compound composition	$Zr_5AlGe_3$
Formula weight	701.02
Structure type	$Hf_5CuSn_3$
Pearson symbol	$hP18$
Space group	$P6_3/mcm$
Unit-cell parameters: <i>a</i> , Å	8.104(3)
<i>c</i> , Å	5.654(2)
Cell volume <i>V</i> , Å <sup>3</sup>	321.62(18)
Formula units per cell <i>Z</i>	2
Density <i>D<sub>x</sub></i> , g cm <sup>-3</sup>	7.241
Absorption coefficient $\mu$ , mm <sup>-1</sup>	22.2
Number of reflections: measured	1558
independent	248
with $F > 3\sigma(F)$	248
Reliability factor $R_{eq}$	0.0392
Range of <i>h</i> , <i>k</i> , <i>l</i>	$-10 \leq h \leq 9, -6 \leq k \leq 12, -8 \leq l \leq 8$
$\theta_{max}$ , °	33.4
Reliability factors: <i>R</i>	0.0357
<i>wR</i>	0.0375
<i>S</i>	1.01

**Table 3** Atomic coordinates, anisotropic and equivalent isotropic displacement parameters for  $Zr_5AlGe_3$  ( $Hf_5CuSn_3$ ,  $hP18$ ,  $P6_3/mcm$ ,  $a = 8.104(3)$ ,  $c = 5.654(2)$  Å,  $R = 0.0357$ ,  $wR = 0.0375$ ).

Site	Wyckoff position	$x$	$y$	$z$	$B_{eq}, \text{Å}^2$
Zr1	6g	0.2567(2)	0	¼	1.21(4)
Zr2	4d	⅓	⅔	0	0.87(3)
Al	2b	0	0	0	1.27(14)
Ge	6g	0.6100(2)	0	¼	1.05(4)

Site	$B_{11}, \text{Å}^2$	$B_{22}, \text{Å}^2$	$B_{33}, \text{Å}^2$	$B_{12}, \text{Å}^2$	$B_{13}, \text{Å}^2$	$B_{23}, \text{Å}^2$
Zr1	1.25(4)	0.87(5)	1.09(5)	0.43(2)	0	0
Zr2	0.86(3)	0.86(3)	0.59(5)	0.43(2)	0	0
Al	1.4(2)	1.4(2)	0.6(2)	0.69(9)	0	0
Ge	0.96(4)	0.94(5)	0.94(5)	0.47(3)	0	0

**Table 4** Interatomic distances ( $\delta$ ) and coordination numbers (CN) in the structure of  $Zr_5AlGe_3$  ( $Hf_5CuSn_3$ ,  $hP18$ ,  $P6_3/mcm$ ,  $a = 8.104(3)$ ,  $c = 5.654(2)$  Å).

Atoms	$\delta, \text{Å}$	CN
Zr1 – 2 Al	2.5151(8)	11
– 1 Ge	2.782(1)	
– 2 Ge	2.864(1)	
– 2 Ge	3.0261(8)	
– 4 Zr2	3.3702(9)	
Zr2 – 2 Zr2	2.8270(8)	14
– 6 Ge	2.8752(8)	
– 6 Zr1	3.3702(9)	

Atoms	$\delta, \text{Å}$	CN
Al – 6 Zr1	2.5151(8)	14
– 2 Al	2.8270(8)	
– 6 Ge	3.4616(9)	
Ge – 2 Zr1	2.782(1)	13
– 2 Zr1	2.864(1)	
– 4 Zr2	2.8752(8)	
– 1 Zr1	3.0261(8)	
– 2 Ge	3.3430(9)	
– 2 Al	3.4616(9)	

The shortest distances in the structure are distances between the sites Zr1 and Al (2.515 Å). These are shorter than in the binary aluminide  $Zr_5Al_4$  (2.849 Å) [12], which crystallizes with a  $Ti_5Ga_4$ -type structure. Insertion of Al atoms into  $Zr1_6$  octahedra in the structure of  $Zr_5Ge_3$  [13] leads to an expansion of these (the distance from the center of the octahedron to its vertices increases from 2.340 Å in  $Zr_5Ge_3$  to 2.515 Å in  $Zr_5AlGe_3$ ) and to an expansion of the hexagonal unit cell. Another feature of the structure of the title compound is the short Zr2-Zr2 distance (2.827 Å), indicating strong interaction between Zr atoms. Similar behavior has been reported for the isostructural phases listed in Table 1.

The content of the unit cell of  $Zr_5AlGe_3$  and the coordination polyhedra of the atoms are shown in Fig. 1. The Zr atoms are coordinated by 11 (site Zr1) and 14 (site Zr2) atoms and form polyhedra of composition  $Zr1Al_2Ge_5Zr_4$  and  $Zr2Ge_6Zr_8$ , respectively. The defect anticubooctahedron  $Zr1Al_2Ge_5Zr_4$  can also be seen as a deformed pentagonal bipyramid of composition  $Al_2Ge_5$  with four additional Zr atoms, or as a tricapped trigonal prism of composition  $Ge_5Zr_4$  with two additional Al atoms. The Frank-Kasper polyhedron  $Zr2Ge_6Zr_8$  can be described as a deformed hexagonal prism of composition  $Ge_6Zr_6$  with two Zr atoms capping the

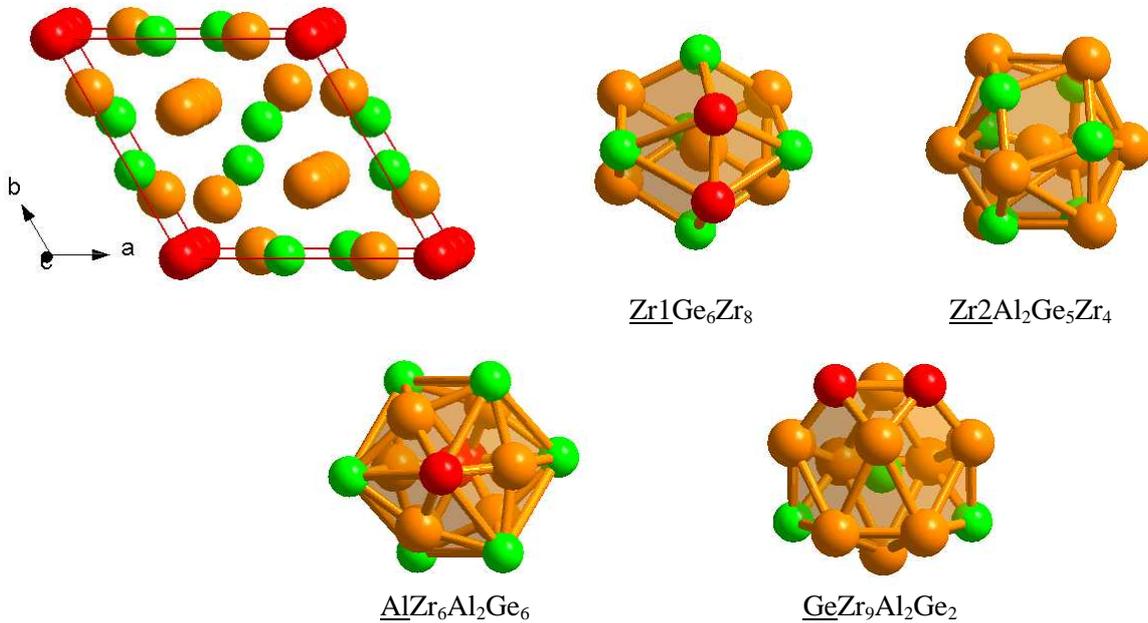
hexagonal faces. The atoms belonging to the closest environment of the Al atoms also form Frank-Kasper polyhedra consisting of 14 atoms,  $AlZr_6Al_2Ge_6$ . These polyhedra are deformed rhombododecahedra formed by an  $Al_2Ge_6$  cube and a  $Zr_6$  octahedron. The Ge atoms are coordinated by nine Zr atoms, two Al and two Ge atoms forming a Frank-Kasper 13-vertex polyhedron of composition  $GeZr_9Ge_2Al_2$ . It can be described as a deformed anticubooctahedron with one additional atom, or as a tricapped trigonal prism of composition  $Zr_9$  with four additional atoms.

The coordination number and polyhedron of the Zr1 site changes when Al is inserted into the structure of the binary compound  $Zr_5Ge_3$ . The 15-vertex polyhedron in the structure of  $Zr_5Ge_3$  (deformed pentagonal prism  $Zr1_0$  with five additional Ge atoms capping the two bases and three lateral faces) transforms into an 11-vertex polyhedron in the structure of  $Zr_5AlGe_3$ . The coordination numbers and polyhedron types of the other atoms do not change.

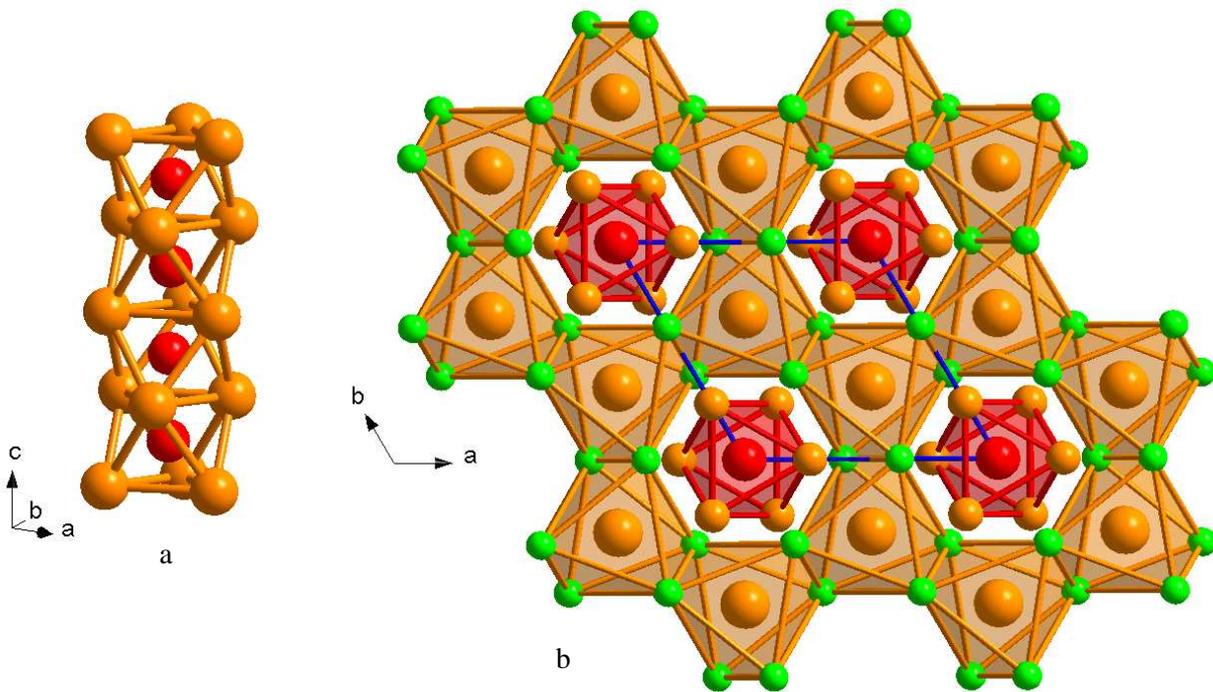
In the structure of the ternary compound  $Zr_5AlGe_3$  columns of simple polyhedra may be considered. Distorted  $ZrGe_6$  octahedra share edges, forming cavities in which  $AlZr_6$  octahedra are situated (Fig. 2). The  $ZrGe_6$  octahedra, as well as the  $AlZr_6$  octahedra, are connected by common faces along the

crystallographic direction  $[001]$  and form a 3D-framework and 1D-rods, respectively. It is interesting to note that, in the structure of  $Zr_5AlGe_3$ , as in the structures of the isotypic ternary phases listed in

**Table 1**, atoms of a  $d$ -element (Zr) center octahedra formed by atoms of a  $p$ -element (Ge) and *vice versa*, Al atoms ( $p$ -element) center octahedra formed by Zr atoms ( $d$ -element).



**Fig. 1** Content of the unit cell and coordination polyhedra of the atoms in the structure of  $Zr_5AlGe_3$  (yellow balls – Zr, red – Al, and green – Ge atoms).



**Fig. 2** Part of an infinite column of Al-centered  $Zr_6$  octahedra along  $[001]$  (a) and 3D-framework of Zr-centered  $Ge_6$  octahedra with Al-centered  $Zr_6$  octahedra in channels (b) in the structure of  $Zr_5AlGe_3$  (yellow balls – Zr, red – Al, and green – Ge atoms).

## Conclusions

A new ternary compound,  $Zr_5AlGe_3$ , was synthesized and its crystal structure was determined by means of X-ray single-crystal diffraction. The structure belongs to the structure type  $Hf_5CuSn_3$ , which is a ternary ordered derivative of  $Ti_5Ga_4$  and a filled-up variant of the  $Mn_5Si_3$  type. The Al atoms occupy Wyckoff position  $2b$  at the origin of the unit cell and center  $Zr_6$  octahedra. The structure can be visualized as built up from columns of  $ZrGe_6$  and  $AlZr_6$  octahedra running along the crystallographic direction [001].

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